New synthesis routes and catalytic applications of ferrierite crystals. Part 1: 1,8-Diaminooctane as a new OSDA

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2 **Part 1: 1,8-diaminooctane as a new OSDA**

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19 **ABSTRACT**

20 *Ferrierite zeolite crystals were successfully prepared by a long amine as 1,8-*

21 *diaminoctane (DAO) as organic structure directing agent. By comparison, shorter*

22 *amine, i.e. ethylenediamine (EN), was also used as OSDA. Results shows that the*

23 *presence of DAO reduces the crystallization time and allows to obtain single FER-type*

24 *crystlas with smaller size and higher acidity, being these significant parameters to take*

25 *into account when zeolites are applied in catalysis. The obtained samples were*

26 *characterized with XRD, TG/DTA, SEM, N_2 adsoprtion/desorption, FTIR and ^{27}Al-*

27 *NMR. In particular, TGA clearly shows that DAO may be accommodate in zeolite*

28 *framework and theoretical modeling suggests a preferential location in the 10-MR ring*

29 *channels.*

30

33 **1. INTRODUCTION**

34 The possibility of preparing zeolites with tailored physicochemical properties by both in

35 situ and post-synthesis treatments is of considerable importance for the industrial

36 perspective, particularly in the area of heterogeneous catalysis where zeolites are widely

37 used [1-6]. In fact, the suitability of zeolite for specific applications, such as catalysis,

38 strongly depend on several factor such as zeolite structure (i.e. size, shape and

39 orientation of the channels), its chemical composition, acidity and crystal morphology.

40 At moment, more than 240 zeolite framework types are known and most of them are not

41 currently applied in industrial processes due to both costs and performances aspects, and

very few zeolite structures have found significant catalytic application in petrochemical and refining industries. FAU (X and Y), MFI (ZSM-5 and Silicalite-1), MOR, BEA (beta), MWW (MCM-22), AEL (SAPO-11) and CHA (SAPO-34) are examples of zeolite structures currently used in industrial catalysis.

Ferrierite (FER) is another zeolite structure that has or may have some importance for industrial applications. FER framework consists of an intersecting 2-dimensional system of 10- (5.4 x 4.2 Å) and 8- (4.8 x 3.5 Å) member ring channels and can by synthesised with Si/Al ratios ranging from 5 to infinity [8, 9]. The particular channel configuration of FER framework is appropriate for processing small molecules. For instance, FER-type zeolite is mainly applied as catalyst in the skeletal isomerization of n-butene for the production of iso-butene and as additive to ZSM-5 catalyst in the dewaxing by selective cracking process for the production of lubricant [10]. Recently, FER-type crystals have also ranked as a highly performing catalyst for dimethyl ether (DME) production via both methanol dehydration reaction and one-pot CO or CO_2 hydrogenation, if compared with other zeolite structures such as MFI, BEA, MOR, CHA, MTW and TON [11-14]; the superiority of ferrierite over the traditional catalyst for DME synthesis, i.e. γ-Al_2O_3, was also reported [15].

Ferrierite can be synthesized combining both Na^+ and K^+ cations [16]. However, it is usually obtained from sodium and OSDA containing systems which give more flexibility in the control of the physicochemical properties of the ultimate product. Amines are often used as OSDA in the synthesis of FER-type materials. Göğebakan et al. [17] reported a study on crystallization field and rate in presence of pyrrolidine or ethylenediamine in the synthesis temperature range 150-225 °C, founding a strong dependence of crystallization rate from temperature, whereas both SiO_2/Al_2O_3 ratio (in

66 the range 10-25) and OSDA amount did not significantly influence the crystallization

67 rate. On the contrary, OSDA affected crystal morphology: individual or platelet

68 crystallites were obtained in presence of pyrrolidine or ethylenediamine, respectively.

69 Furthermore, the authors even found mordenite as competitive phase only when

70 pyrrolidine is used as OSDA.

71 Kamimura et al. [9] reported the synthesis of ferrierite in presence of pyridine as

72 OSDA, obtaining well-defined plate-like crystals with a SiO_2/Al_2O_3 up to 324. In

73 particular, the synthesis of pure siliceous ferrierite was successful in presence of sodium

74 fluoride. Aluminium-free ferrierite was also obtained in presence of ethylediamine as

75 OSDA and boric acid as templates [18].

76 On the contrary, Forbes et al. [19] reported that, using diethanolamine as OSDA, pure

77 phase ferrierite crystals may be obtained only in a narrow Si/Al ratio. In particular, the

78 authors found that by increasing the SiO_2/Al_2O_3 from 16 to about 25, a ferrierite/ZSM-5

79 mixed phase was obtained.

80 OSDA type as well as synthesis environment also affects the morphology and the size

81 of the final crystals. In this concern, Rakozy et al. [20, 21] report the synthesis of all-

82 silica large FER-type crystals by using unbranched monoalkyl C3-C5 amines in mixture

83 with pyridine, showing the possibility to tune the shape and size of obtained crystals. In

84 particular, by increasing the number of carbon atoms of the amine, the crystal shape

85 changed from hexagonal to rectangular flat prismatic morphology.

86 Lee et al. [22] successfully synthesised ferrierite zeolite nanoneedles about 10 nm in

87 diameter and 100 nm in length. The success of the synthesis strongly depends on both

88 Si/Al ratio and ion types. In particular, the authors obtained high pure ferrierite by

4

89 adopting a Si/Al=20 in presence of Na+, while mixed or amorphous phases were

90 obtained for lower or higher aluminium content and using other ion types (e.g. Li^+, K^+,

91 Rb^+, Cs^+).

92 The effect of organic solvent such as triethylamine, tetra- or diethylene glycole,

93 dimethylsulfoxide, lutidine and mineral oil on the shape and size of final crystals from

94 non-aqueous syntheses was also deeply investigated [23].

95 Pinar et al. assert that the choice of organic template molecules (OSDA) in FER-type

96 materials is of paramount importance for aluminium distribution and incorporation [23],

97 affecting acid sites distribution, location, and accessibility [24]. The authors synthesised

98 ferrierite crystals with a Si/Al ratio of about 15 starting from a sodium-free gels in

99 fluoride media by using some specific combinations of tetramethylammonium (TMA),

100 pyrrolidine and benzylmethylpyrrolidinium (bmp), founding that the acid sites

101 distribution between the ferrierite cage accessible to through 8-MR window and the 10-

102 MR pores depend on the OSDA mixture composition. In particular, the accessibility to

103 Brønsted sites is higher when TMA/bmp or TMA/pyrrolidine mixture are used

104 respected to alone pyrrolidine with a direct effect on catalysis. Therefore, the already

105 published studies highlighted the possibility to obtain tailored ferrierite crystals by

106 choosing proper SDAs and synthesis conditions, with an effective control of both acid

107 properties and crystal morphology.

108 In this paper, FER-type crystals were synthesized by using a long amine as 1,8-

109 diaminooctane as OSDA. This molecule is usually used for the synthesis of MEL, TON

110 and ZSM-48 zeolites. In the case of MEL and TON zeolite, the success of the

111 crystallization of the desired phase strongly depends on the stirring condition: static or

112 high-speed agitation condition lead to the selective formation of high pure MEL or TON

113 phase, respectively [25]. Crystallization with stirring is also required for the synthesis of

114 ZSM-48 [26].

115 To the best of our knowledge, there has been no report on the synthesis of ferrierite by

116 using 1,8-diaminoctane as OSDA. In this work the crystallization of ferrierite in

117 presence of 1,8-diaminioctane is carried out with the aim to assess the effect of this

118 molecule on physico-chemical properties of obtained materials. Furthermore, in order to

119 have new insights about the role of amine molecule length, FER-type zeolite with

120 ethylenediammine as OSDA was also prepared for comparison.

121 TG/DTA analysis allowed to assess the interaction between the OSDA and zeolite

122 framework whilst DFT calculations gave information about molecules location.

123 Brønsted/Lewis acid sites distribution and concentration of synthesised materials have

124 been investigated with NH_3-TPD and FT-IR of several adsorbed probe molecules while

125 morphology and crystallinity has been analysed by scanning electron microscopy and

126 powder XRD, respectively. Finally, DFT calculations allowed validate the experimental

127 evidences.

128

129 **2. EXPERIMENTAL**

130 *2.1 Samples preparation*

131 The following chemicals were used in all the synthesis: colloidal silica Ludox AS-40

132 (SiO_2, 40 wt% suspension in H_2O, Aldrich), sodium aluminate ($NaAlO_2$, Aldrich),

133 sodium hydroxide (pellets, Aldrich), ethylenediamine (Fluka), 1,8-diaminooctane

134 (Aldrich), and distilled water.

135 Synthesis of FER sample by using ethylenediamine (En-FER) was prepared by adopting

136 the following synthesis gel molar composition [28]:

137 $$19.7 \, En - 1.85 \, Na_2O - 15.2 \, SiO_2 - 1 \, Al_2O_3 - 590 \, H_2O$$

138 In a typical synthesis 0.3 g of NaOH and 1.3 g of sodium aluminate were dissolved in

139 64.1 g of distilled water. Then 8.2 g of ethylenediamine were added to dropwise and the

140 solution was stirred for 30 min. Afterwards 15.8 g of colloidal silica were added

141 dropwise to the solution, and the obtained gel was stirred (300 rpm) at room

142 temperature for 1 h. Crystallization was carried out in 120 ml stainless-steel Teflon

143 coated autoclave up to 12 days at 180 °C in a tumbling oven (rotation speed: 15 rpm).

144 The potential role of 1,8-diaminooctane as OSDA was investigated by gradually

145 replacing En with DAO, starting from the En-FER (i.e. DAO free) synthesis gel. FER-

146 type crystals were obtained from synthesis gel with different DAO/En molar ratio: 0.25,

147 0.5, 1, 2 and ∞ (i.e. En free), by using the following molar gel composition

148 $$x \, DAO - 1.85 \, Na_2O - 15.2 \, SiO_2 - 1 \, Al_2O_3 - 590 \, H_2O - y \, En$$

149 with $5<x<10$ and $0<y<19.7$.

150 In the synthesis of DAO-FER only DAO was used as SDA. For this latter synthesis, 0.3

151 g of NaOH and 1.3 g of sodium aluminate were dissolved in 21 g of distilled water. A

152 secondary solution was prepared by dissolving 7.8 g of DAO into 42 g of distilled

153 water, under vigorous stirring. Afterwards, the two solutions were blended and 15.8 g of

154 colloidal silica were added dropwise and the obtained gel was stirred (300 rpm) at room

155 temperature for 1 h. Crystallization was carried out in 120 ml stainless-steel Teflon

156 coated autoclave up to 12 days at 180 °C in a tumbling oven (rotation speed: 15 rpm).

After crystallization, the solid was separated by vacuum filtration and washed several times with distilled water until the pH of filtered solution was close to the neutral value. The obtained white solid was then dried for 15 hours in a static oven kept at 80 °C. In order to remove template from zeolite, air-flow calcination was carried out at 550 °C with an holding time of 8 hours and a heating rate of 2 °C/min. Afterwards, H-form solid was obtained by ion-exchange with a NH_4Cl 1 M solution at 80 °C followed by a secondary air-flow calcination at 550 °C with an holding time of 8 hours and a heating rate of 2 °C/min [29].

2.2 Physicochemical characterization

X-Ray powder diffraction (APD 2000 Pro) (region 5° < 2θ < 50°, step 0.02°/ s) was used to verify both purity and crystallinity of obtained phase; the morphology of the crystalline phase was observed on a scanning electron microscope (SEM, FEI model Inspect) and a transmission electron microscopy (TEM, Philips CM12). The sodium, aluminum and silicon content in the calcined catalysts was measured by atomic absorption spectroscopy (GBC 932 AA).

The specific surface area and the micropores volume of the calcined samples were obtained from BET and t-plot analysis of porosimetry data (ASAP 2020 Micromeritics) under nitrogen adsorption at -196 °C, after a pre-treatment in vacuum at 200°C for 12 h. The thermo-gravimetric analysis over the as-synthesised samples was performed on the automatic TG/DTA instrument (Shimadzu) under 50 cc/min of air flow in the temperature range 25-850°C (heating rate of 5 °C·min^{-1}).

Solid-state magic angle spinning nuclear magnetic resonance spectroscopy (MAS-NMR) was used to characterize the local ^{27}Al environment in the zeolite samples. All data were recorded on a Bruker Advance 500 MHZ spectrometer using a 4.0 mm rotors.

181   The ^{27}Al-NMR spectra were acquired at 130.3 MHz with radiofrequency power of 54

182   kHz, a pulse length of 4.6 μs with a corresponding angle theta of π/4. Rotors were spun

183   at 14 kHz, and the recycle delay used for both nuclei was 1 second.

184   Surface acidity of H-form sample was measured by NH$_3$-TPD analysis (TPDRO1100,

185   ThermoFisher) according to the following procedure. Dried sample (100 mg, pellet

186   mesh 90-150 μm) was loaded in a linear quartz micro-reactor and pre-treated at 300°C

187   in helium flow for 1 h to remove adsorbed water. The sample was cooled down to

188   150°C and saturated with 10% v/v NH$_3$/He mixture with a flow rate of 20 STP mL·min^{-}

189   ^{1} for 2 h. Ammonia physically adsorbed was removed by purging in helium at 150°C for

190   1 h until TCD baseline stabilization. Desorption measurement was carried out in the

191   temperature range of 100-950°C (10 °C·min^{-1}) using a helium flow rate of 20 STP

192   mL·min^{-1}.

193   Brønsted and Lewis acid sites concentration was measured via FT-IR analysis by using

194   D$_3$−acetonitrile as a probe molecule. Infrared spectra were recorded with a Bruker

195   Equinox 55 spectrometer. The adsorption of D$_3$−acetonitrile was performed at room

196   temperature with a pressure of 10 Torr, after that the samples were degassed under high

197   vacuum condition (cell pressure: 10^{-6} Torr) at 300°C for 4 h, in order to purge adsorbed

198   water. Physisorbed acetonitrile was removed by further degassing until the vacuum

199   condition was recovered. The FT-IR spectra were recorded in the temperature range 25-

200   300 °C (temperature step of 50°C), outgassing the sample for 30 minutes at any

201   temperature variation.

202   The amount of D$_3$−acetontrile adsorbed on Brønsted and Lewis was determined by

203   using a literature value for the molar absorption coefficient [30]. The number of

204   Brønsted sites was calculated from the area of the bands observed at 2297 cm^{-1} by

205   adopting an extinction coefficient of 2.05 cm/µmol, whilst the number of Lewis acid

206   sites was calculated from the area of the bands between 2310–2325 cm^{-1} by adopting an

207   extinction factor of 3.6 cm/µmol. Peaks analysis and deconvolution for both FT−IR and

208   NH_3-TPD profiles were performed by using commercial software (PeakFit 4.12,

209   Seasolve−USA).

210

211   **3. RESULTS AND DISCUSSION**

212   *3.1 Structural properties of FER-type zeolites synthesized with different OSDA*

213   X-ray patterns as a function of crystallization time of En-FER and DAO-FER are

214   reported in Figure 1. Crystal growth kinetics of En-FER is relatively slow (see Figure 1-

215   A) as traces of FER phase are observed after 3 days and after 6 days the crystalline

216   phase is dominant. A crystallization time of 10 days can be considered as optimal to

217   obtain high crystalline En-FER since, for longer time (12 days), traces of kaetite phase

218   appear. Figure 1-B reports the XRD patters as a function of crystallization time for

219   DAO-FER sample. Results show that En replacing with DAO speeds the crystallization.

220   In fact, after 3 days FER-phase is almost completely crystallized whilst only traces of

221   crystalline product are observed for En-FER after the same synthesis period. That

222   evidence suggests that in presence of DAO, the induction period is shortened, even

223   reducing the crystal size as discussed later.

224   Because DAO was never reported as a template in FER zeolite synthesis, a deeper

225   investigation by TG/DTA analysis of as-synthesised materials was performed. The

226   thermo-analytic results of DAO-containing systems are reported in Figure 2, in terms of

227   thermogravimetric (TG) and first derivate (dTG) profiles. Heat flow profiles (DTA) are

228   reported in Figure S.1 (Supporting Information).

229   From the analysis of the dTG profiles, it can be deducted that 1,8-diaminooctane might

230   plays the role of template for FER structure. In fact, dTG profile of DAO-free sample

231   (En-FER, Figure 2A) shows the presence of four main peaks [28]. The endothermic

232   peak (see Figure S.1A) at ca. 73 °C is attributed to water release and the corresponding

233   weight loss for En-FER was of ca. 2 wt%. In addition, En-FER sample contains about

234   11 wt% of ethylendiammine, released in the temperature range 250-700 °C. The

235   associated exothermic phenomena (see DTA profile reported in Figure S.1A) are related

236   to the organic combustion. It is evident from Figure 1A, that at least four different dTG

237   peaks can be associated to ethylenediamine decomposition (249°C, 302 °C, 420 °C and

238   605 °C), indicating the presence of different interactions between OSDA and zeolite

239   framework. As the DAO/En ratio increases from 0 to 0.5 or 1 an effect of a shift toward

240   low temperatures is observed for these dTG peaks. If the peak of En-FER at 420° C is

241   considered, it clearly appears that it decreases down to 366 °C and 352 °C (Figures 2B-

242   C) when the DAO/En ratio is increased to 0.5 or 1, respectively. Also, dTG profile

243   reported in Figure 2C shows that the interaction between ethylenediamine and zeolite

244   framework characterised by the lowest temperature decomposition (around 249°C for

245   En-FER) tends to disappear when DAO progressively replaces En as OSDA. It is

246   worthy to clarify that this effect it is not attributable to lower amount of En in synthesis

247   gel because the TG/dTG profile of a DAO-free FER synthesised with the same

248   ethylenediamine content of the sample with DAO/En=2 (i.e. $En/Al_2=5$) shows the same

249   peaks of the En-FER ($En/Al_2=19.7$), as reported in Figure S.2 of Supporting

250   information. This result suggests that the amount of ethylediamine in the synthesis gel

251   slightly affects its incorporation and the changing of dTG profiles that are, on the

252   contrary, mainly affected by the presence of DAO in the synthesis gel and then in the

253   zeolite structure. The effect of the only DAO in the FER synthesis is shown in Figure

254   2.D, where the main exothermic peak at 348 °C can be unequivocally attributed to 1,8-

255   diaminooctane combustion. The weight loss due to organic molecules decomposition,

256   measured in the temperature range 250-650 °C decreases from ca. 11 wt% to ca. 7 wt%

257   when the DAO/En ratio increases from 0 to ∞ respectively. On the contrary, the weight

258   loss between 25 and 200 °C, attributed to physically adsorbed water, increase from 2

259   wt% to 5 wt % by increasing the DAO/En ratio from 0 to ∞, respectively, suggesting

260   that the presence of 1,8-diaminooctane also promotes the water incorporation. Even

261   though the role and the location of DAO molecules is not fully elucidated yet, we

262   anticipate that DAO molecules are located inside the 10 MR of FER structure since the

263   8 MR channel is too small to accommodate them.

264   Figure S.3 shows the ^{27}Al MAS NMR spectra of the H-form FER-type zeolites with En

265   or DAO as OSDA. They are dominated by a signal centered at around 55 ppm,

266   characteristic of tetra-coordinated aluminium species in zeolite framework. A broad

267   signal at around 0 ppm revealing the presence of octahedral extra-framework aluminium

268   (about 10%) for both the samples. The observed quadrupolar line broadening can be

269   caused by distortions of the octahedral symmetry of AlO_6 sites.

270   It is well-known that the zeolite structure is generated by a network of SiO_4 and AlO_4^-

271   tetrahedral linked together by a shared oxygen atom. Due to the presence of trivalent

272   atoms (e.g. aluminium) in the tetrahedral units, the framework offers a negative charge

273   which needs to be balanced with non-framework exchangeable cations that are

274   generally elements of the group IA and group IIA as sodium, potassium, magnesium

275 and calcium. In this work, sodium is used to synthesise FER-type materials, and this is

276 the cation that balances negative charges associated with aluminium atoms.

277 Furthermore, the high quantity of tetrahedral aluminium species detected by NMR

278 analysis, suggests that the Na/Al ratio should be theoretically around one. The chemical

279 analysis reported in Table 1 shows that Na/Al is 0.06 for En-FER and 0.6 for DAO-

280 FER. These results disagree with the previous theoretical assumption, suggesting that

281 sodium cations do not balance an important amount of aluminium species in tetrahedral

282 position, expecially. Vuono et al. [31] reported a Na/Al ratio value below to unity

283 (around 0.6) for MCM-49 materials synthesised in the presence of hexamethylenimine

284 despite aluminium was mainly tetra-coordinated as in our case. Forbes et al. [32, 33]

285 report that during the synthesis of ZSM-5/Theta-1 materials in the presence of

286 diethanolammine, the organic molecules were considered to have both a pore-filling

287 role and charge compensatory role. Furthermore, Rollmann et al. [34] assert that

288 protonated amine (e.g. ethylamine, pyrrolidine, hexamethylenimine) play the role as

289 counter ion of negative charge associated to tetrahedral aluminium atoms located into

290 zeolite framework, also for FER-type materials. Therefore, because in this work,

291 aluminium atoms are mainly present in tetrahedral coordination, protonated organic

292 molecules can be reasonably present to compensate negative charge of zeolite lattice.

293 Therefore, theoretically the $(OSDA^{+}+Na^{+})/AlO_2^{-}$ molar ration should be about one in

294 order to have the balance of aluminium negative charge. In this regard, it is possible to

295 calculate the unit cell molar composition for En-FER and DAO-FER, respectively, as it

296 follows:

297 En-FER: $En_{3.9}Na_{0.2}Al_{3.8}Si_{32.2}O_{72} \cdot 2.8\ H_2O$

298 DAO-FER: $DAO_{1.0}\ Na_{2.8}Al_{4.7}Si_{31.3}O_{72} \cdot 7.8\ H_2O$

Results clearly show that ethylenediamine is able to compensate the aluminium negative

charge of zeolite framework since En^+/AlO_2^- molar ration is near one and a negligible

sodium amount is incorporated. On the contrary, 1,8-diaminooctane is not able to

compensate all of the aluminium negative charge and a higher amount of sodium is

incorporated. Moreover, in order to respect the electro-neutrality of the framework, one

DAO molecule should be able to compensate two aluminium negative charge. This

condition may be obtained if both the amino groups of DAO molecules are protonated

and able to compensate two different aluminium negative charge. This statement is also

confirmed by DFT theoretical study later reported.

The results of N_2 adsorption analysis are summarized in Table 1. En-FER sample shows

a slightly higher micropore volume than DAO-FER sample. This difference may be

related to the sodium content.

Figure 3 shows the adsorption/desorption isotherms of synthesised samples in protonic

form. Both En-FER and DAO-FER sample exhibit a type I adsorption isotherms typical

of microporous materials [28] but some differences may be highlighted. In fact, for En-

FER sample an uptake increase is observed at P/P°>0.9, that is less evident for DAO-

FER sample. This isotherm shape of En-FER may indicate the presence of external

mesopores due to agglomeration of crystals, as also revealed by SEM.

Figure 4 shows representative SEM micrographs reveailing that all samples exhibit the

typical of ferrierite plat-like morphology with a well-developed {1 0 0}. However, the

crystals differ in size and the level of agglomeration. The En-FER crystals (Figure 6A)

have the following characteristics: length 2-3 μm, length/width ratio around 2 and

thickness 100 nm, while DAO as OSDA generates smaller crystals with size below 1

μm and lower length/width ratio (around 1.3). As showed by XRD analysis, a higher

323 crystallization rate is observed for DAO-FER sample, that may cause the formation of

324 smaller crystals. Moreover, a significant agglomeration is observed for En-FER sample,

325 while single crystal is obtained in presence of DAO.

326 *3.2 Zeolite acidity*

327 The NH_3-TPD profiles are reported in Figure 5 for all the synthesised samples

328 displaying three main peaks characterised by different ammonia maximum desorption

329 temperatures peaks (T_M) as an indicator of different acid sites family (the higher is T_M,

330 the stronger is the acid sites).

331 As suggested by Niwa and Katada [35, 36], the interpretation of peaks with T_M<300 °C

332 may be quite misleading since the desorption effect may also be related to physically

333 adsorbed ammonia interacting via a hydrogen bond with the NH_4^+ ions adsorbed onto

334 acid sites. In this condition, having comparable desorption energy, the presence of weak

335 acid sites, may be partially hidden by physisorbed ammonia. On the contrary, the peaks

336 with 300°C<T_M<500 °C are associated with ammonia molecules desorbed from strong

337 acid sites (both Brønsted and Lewis acid sites). The peak with T_M>700 °C can be

338 associated to dehydroxylation phenomena [37, 38] since it is detected either in NH_3-

339 TPD experiments or in the corresponding "blank" TPD experiments performed over the

340 same catalyst and under the same experimental conditions, without NH_3 feed.

341 Quantitative results of NH_3-TPD measurements reported in Table 2. DAO-FER sample

342 exhibits a higher total acid sites concentration according to the higher aluminium

343 content measured by atomic absorption technique. DAO-FER also shows a higher $T_{M,HT}$

344 indicating a major strength of acid sites. Such aspect may be related to aluminium

15

345 distribution and location and further investigations should be done in order to elucidate

346 the role of OSDA on the aluminium siting [39, 40].

347 Figure 6 depicts the FT-IR spectra of H-ferrierite samples before and after adsorption of

348 deuterated acetonitrile. Fore bare samples, two main bands observed at 3745 cm^{-1} and

349 3600 cm^{-1} are associated with terminal silanol Si-OH and Brønsted acid sites Al-OH-Si

350 groups, respectively [41, 42]. A shoulder at around 3645 cm^{-1} is also present. Similar

351 result was reported already [28, 43], but the peak is not well-identified. Peixoto et al.

352 [44] suppose that the observed shoulder can be associated to a silanol group anchored

353 on extra-framework aluminium species (e.g., octahedral aluminium) as reported also by

354 Rachwalik et al. [45] even though in the investigated samples, NMR analysis suggests

355 that octahedral aluminium species are nearly absent. It is known from the literature that

356 aluminium in highly distorted coordination might become NMR silent and the observed

357 band at 3645 cm^{-1} can be associated to OH groups connected to the so called 'invisible'

358 aluminium species [46, 47] that can play a role of Lewis acid sites [48, 49]. The

359 presence of these species could explain the non-stoichiometric Na/Al ratio discussed

360 above.

361 After adsorption of CD$_3$CN, two main bands at about 2296 cm^{-1} and 2322 cm^{-1} appear,

362 revealing the presence of both Brønsted and Lewis acid sites, respectively, in boith cthe

363 samples. Smaller bands at lower wave length associated with physisorbed acetonitrile

364 are also presents [30]. The concentration of Brønsted and Lewis sites were obtained

365 from the integral intensities of the IR bands of adsorbed CD$_3$CN at 2296 cm^{-1} and 2322

366 cm^{-1}, respectively, by using the extinction factor coefficients reported in Section 2.

367 Brønsted and Lewis site distribution of the investigated samples was calculated from the

368 area of bands recorded at 25 °C; both the samples exhibits a significant amount

369 (asround 40 %) of Lewis acid sites.

370 In order to estimate the strength of Brønsted and Lewis acid sites TPD measurements

371 were carried out in the range 25-400 °C. Figure 7 shows both the FT-IR profiles and

372 the fraction of Brønsted and Lewis acid sites free of adsorbed D_3CN molecules as a

373 function of desorption temperature.

374 The set of data show that Lewis acid sites are stronger than Brønsted ones for all the

375 samples. The desorption of Brønsted sites becomes significant above 150°C while the

376 Lewis sites are still occupied above 300°C. In particular, the desorption of acetonitrile

377 molecules starts at 50 °C and no occupied sites are present above 250 °C for both the

378 samples, indicating a similar strength distribution. On the contrary, the desorption of

379 probe molecules from Lewis acid sites starts above 25 °C and 150 °C for En-FER and

380 DAO-FER, respectively. This result indicates that En-FER possesses weaker Lewis acid

381 sites than DAO-FER. By considering as strong Lewis acid sites fraction, the value of

382 Lewis acid sites still occupied by CD_3CN molecules at 300 °C, it is possible to calculate

383 the fraction of strong Lewis sites reported in Table 2 (last column). DAO-FER

384 possesses a higher amount of strong Lewis acid sites than En-FER, indicating that the

385 synthesis system affects both typology and strength of acid sites.

386 *3.3 DTF calculations*

387 As previously discussed, 1,8-diaminooctane was never reported as a template for FER-

388 type structure, to the best of our knowledge. Our data unambiguously show that DAO is

389 incorporated in FER channels and fulfil the function of OSDA and charge balancing

390 cation, similarly to the other nitrogen-SDA. This aspect was also addressed from a

391   theoretical point of view, by performing a DFT simulation for the investigated systems.

392   In this regard, zeolite is considered as an infinite crystal and dispersion corrections are

393   used in order to take into account confinement effects. Both binding energies and length

394   of H(SDA)---O(FER) hydrogen bonds are reported in Table 3 and the optimized

395   structures are displayed in Figure 8.

396   The binding of both En and DAO with the 10 MR pore walls of zeolite are

397   thermodynamically favoured (i.e. negative binding energies are calculated) suggesting

398   that SDAs bind though hydrogen-bond formations with the lattice of zeolite. In

399   particular, the strong interaction energy calculated for En-FER system may be attributed

400   to the short hydrogen bond lengths, that are also almost linear and therefore stronger

401   than the other systems.

402   Similarly, DAO is stabilized in FER cavity via several short and linear N-H---O(ZEO)

403   H-bonds, which explains, also in this case, the computed strong interaction energy with

404   pore walls of FER zeolite. Furthermore, the computed dispersion energies for all the

405   systems is about 0.3% to the total energy, indicating that the is not a particular

406   stabilization of SDA molecules due to the attractive dispersion interactions or

407   confinement effect. Concerning DAO-FER system, DFT calculations suggest that DAO

408   molecules may be located in 10 MR channels of FER zeolite and that the formation of

409   H-bond is the driving force for the stabilization of the molecules within the voids of

410   zeolite. These results are in agreement with the chemical analysis. In fact, in the case of

411   En, SDA molecules are protonated and they are able to compensate the negative charge

412   of aluminium located in both 8 MR and 10 MR channels, and therefore, a low amount

413   of sodium is required. On the contrary, if the accommodation of protonated SDA

414   molecules within 8 MR channels is difficult, as in the case of DAO, more sodium is

415   required for compensating negative charge inside 8 MR channels. This conclusion is in

416   agreement with previous studies on ferrierite [50, 51].

417

418   **4. CONCLUSIONS**

419   In this work, for the first time, FER-type crystals were successfully synthesised by

420   using 1,8-diaminooctane (DAO) as organic structure directing agent (DAO-FER

421   sample). The obtained results in terms of crystals morphology and acidity were

422   discussed by taking FER zeolite prepared with ethylenediamine as reference (En-FER

423   sample). XRD analysis suggests that the presence of DAO reduces the induction time

424   respect to ethylenediamine, leading to the formation of smaller crystals with a lower

425   agglomeration. TG/DTA clearly indicates the presence of DAO inside the cavity of

426   zeolite, whilst chemical analysis suggest that protonated DAO is able to balance AlO_2^-

427   charge of the framework along 10 MR channels, while sodium shall be incorporated

428   along the 8 MR. On the contrary, a negligible amount of sodium is requested in the case

429   of ethylenediamine. In fact, DFT simulations show that DAO may be incorporated

430   along the 10 MR only, whilst ethylenediamine is able to balance the charges of Al

431   centres along both 8 and 10 MR channels. Furthermore, a higher amount of aluminium

432   is incorporated when DAO is used as OSDA leading to a higher total acidity. FT-IR

433   measurements reveal that both DAO- and En-FER samples possess a high amount of

434   Lewis acid sites (about 40%) that are stronger for DAO-FER sample.

435   The obtained results show the possibility to control the physicochemical properties of

436   zeolite crystals by selecting the type of structure directing agent leading to tailored

437   materials for specific applications, e.g. heterogeneous catalysis.

438

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529

FIGURES CAPTIONS

539   **Figure 1**   XRD pattern of FER samples synthesized with ethylenediamine and 1,8-

540   diaminooctane (DAO). (*****) indicates the presence of Kaetite phase.

541   **Figure 2**   TGA (continuous line) and dTG (dashed line) data of FER samples

542   synthesised with DAO/En=0 (A), DAO/En=0.5 (B), DAO/En=2 (C) and

543   DAO/En=∞ (D).

544   **Figure 3** N_2 adsorption (closed circle) and desorption (open circle) isotherms at 77

545   of En-FER and DAO-FER samples.

546   **Figure 4**   SEM micrographs of FER-type materials synthesised with

547   ethylenediammine and 1,8-diaminooctane.

548   **Figure 5** NH_3-TPD profiles of En-FER and DAO-FER samples.

549   **Figure 6**   FT-IR spectra of of En-FER and DAO-FER samples after evacuation at

550   300 °C for 4h (a) and after adoption of CD_3CN molecules at 25°C (b).

551   All the spectra are recorded at 10^{-6} torr.

552   **Figure 7**   FT-IR spectra of D_3-acetonitrile adsorbed on En-FER (A), and DAO-

553   FER (B) evacuated at 25 °C (a), 50 °C (b), 100 °C (c), 150 °C (d), 200

554   °C (e), 250 °C (f) and 300 °C (g). Fraction of Brønsted (●) and Lewis (■)

555   acid sites free of adsorbed D_3CN molecules as a function of desorption

556   temperature for En-FER (A') and DAO-FER(B').

557   **Figure 8**   DFT optimized crystal structures of FER with SDA molecules. SDA in

558   10 MR and in 8 MR are shown in the top and down panels, respectively,

559   in Figs. a, b, c. Optimized unit cells structures of every FER-SDA are

560   displaced at the left-side and 2x2x2 cells - at the right side in Figs. a, b, c

561   and d. The dashed lines indicate all the H-bonds ≤ 3 Å. The lengths of

562   the shorter and linear H-bonds bellow 2.5 Å are reported in Table X.

563   Atomic colour code: dark blue for Si; red for O; light blue for $Al3^{+}$, black

564   for C, violet for N, grey for H and yellow for Na.

565

SAMPLE	En-FER	DAO-FER
Na/Al_{bulk}[a] [mol/mol]	0.06	0.60
Si/Al_{bulk}[a] [mol/mol]	8.4	6.6
S_{BET}[b] [m^2/g]	329	287
V_{mic}[c] [cm^3/g]	0.131	0.125

[a] Atomic absorption

[b] BET superficial area

[c] Micropore volume calculated by *t*-plot method

Table 1

	En-FER	DAO-FER
NH_3-uptake [μmol/g_{cat}]	1052	1418
$T_{M,LT}{}^{a}$ [°C]	245	231
$x_{LT}{}^{[b]}$	0.30	0.40
$T_{M,HT}{}^{c}$ [°C]	436	455
$x_{HT}{}^{d}$	0.70	0.60
Lewis acid sites[e] [%]	0.41	0.40
Strong Lewis acid sites[f] [%]	0.16	0.24

[a] Temperature of maximum desorption of NH_3 between 100 and 300°C

[b] Fractional population of sites between 100 and 300°C

[c] Temperature of maximum desorption of NH_3 above 300°C

[d] Fractional population of sites above 300°C

[e] Estimated by FT-IR analysis of adsorbed D_3-acetonitrile at 25°C

[f] Estimated by FT-IR analysis of adsorbed D_3-acetonitrile at 300°C

Table 2

Sample	Binding energy (eV/mol)		Binding lengths (Å)	
	8 MR	10 MR	8 MR	10 MR
En-FER	-24.01	-25.76	1.820 1.908 1.710	1.864 1.902 2.473
DAO-FER	N.a.N	-22.67	-	2x1.960; 2.042[a] 2.129; 1.498[b] 2.441, 2.488[c]

[a] NH_3^+---O(ZEO) H-bond;
[b] NH_2---O(ZEO) H-bond
[c] C(CH)---O(ZEO) H-bond

Table 3
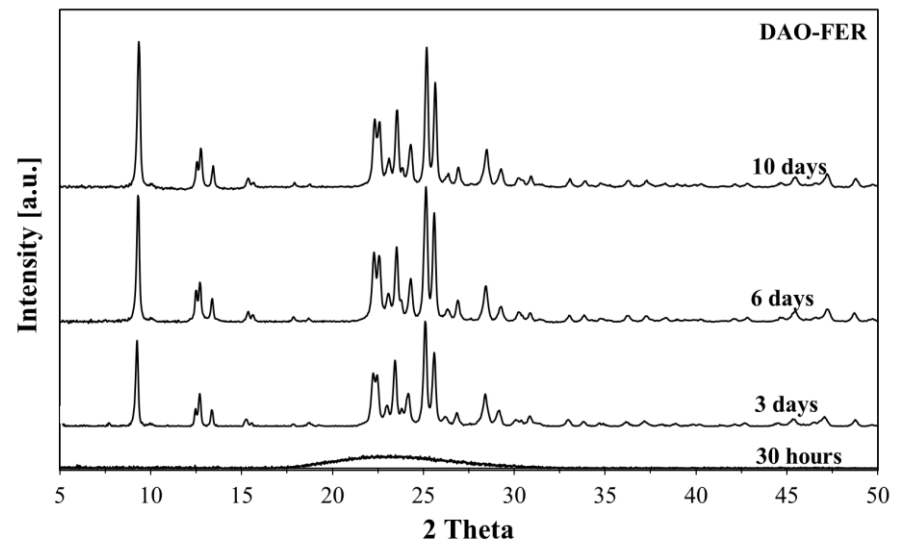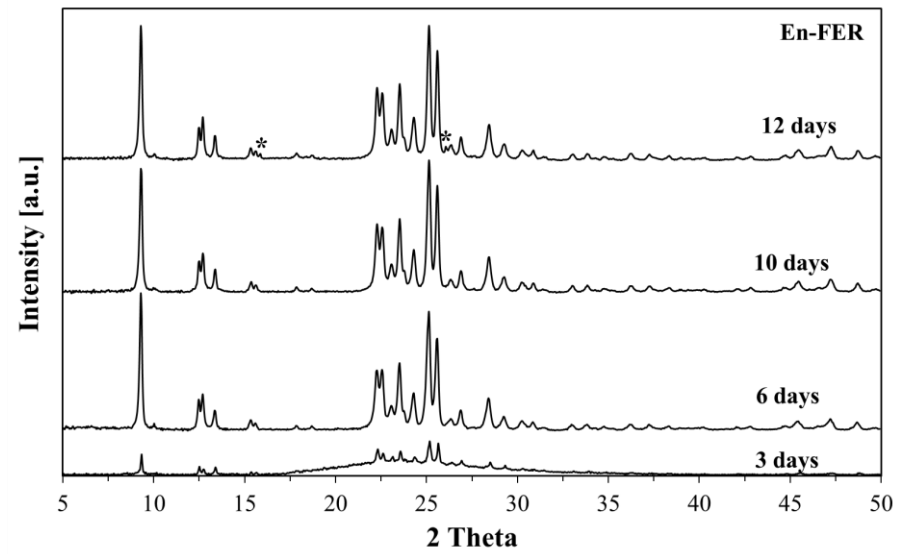
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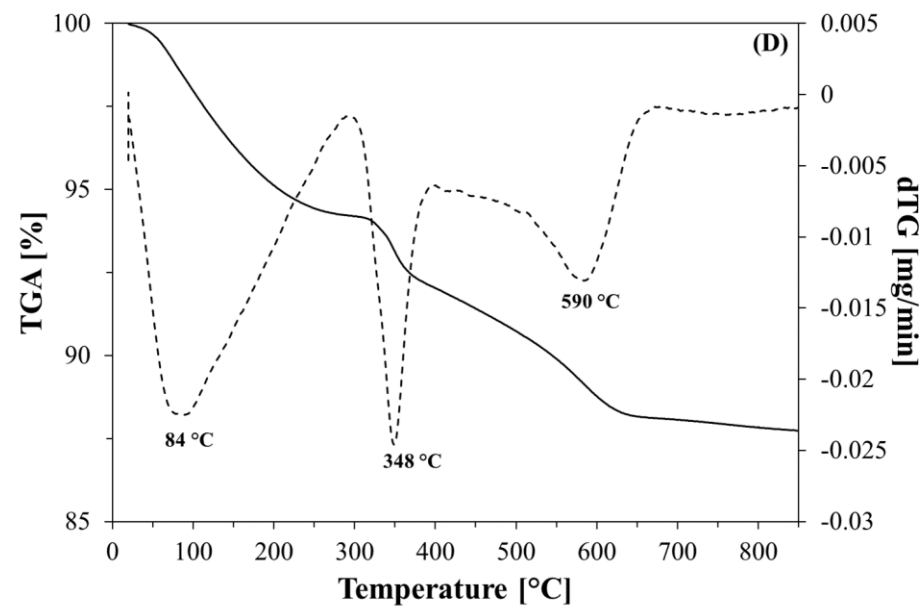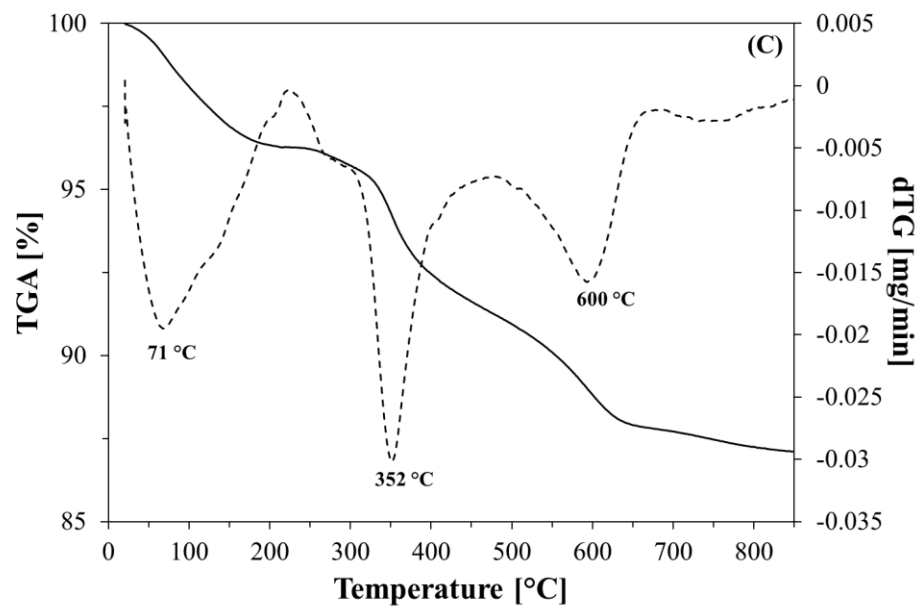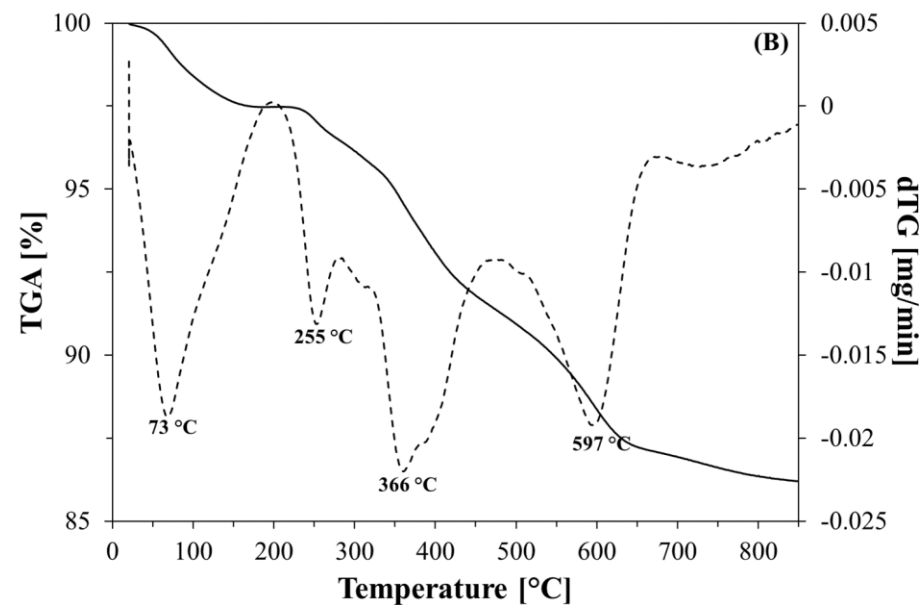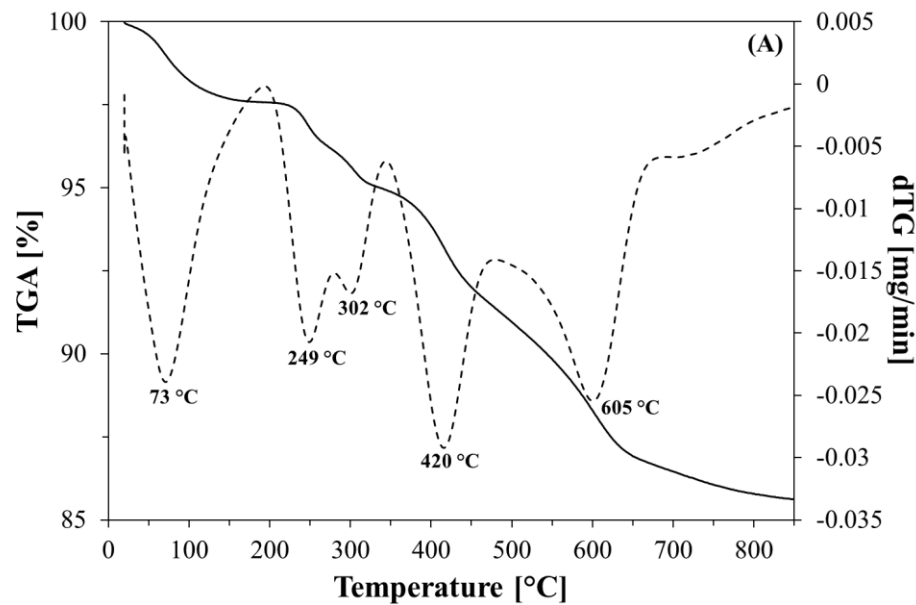
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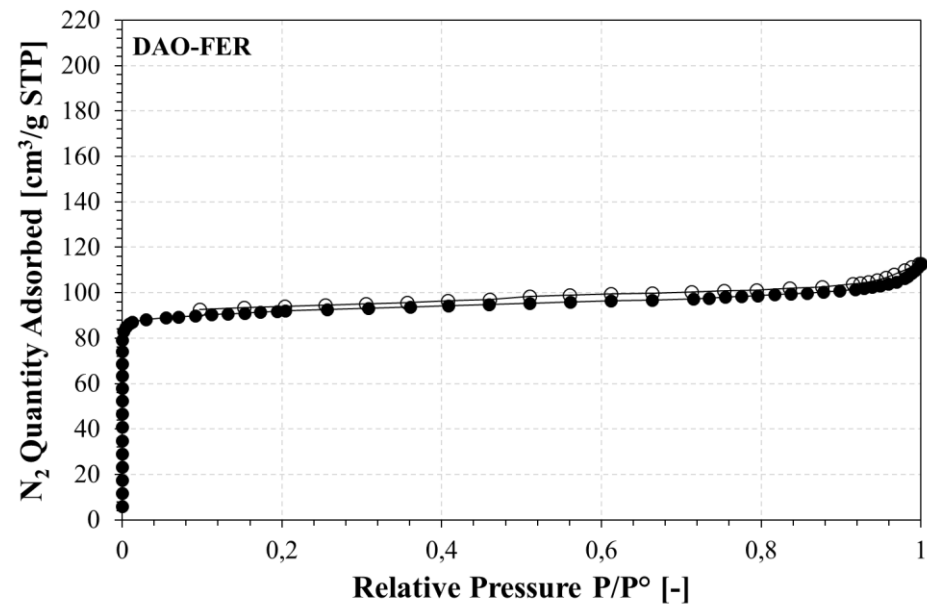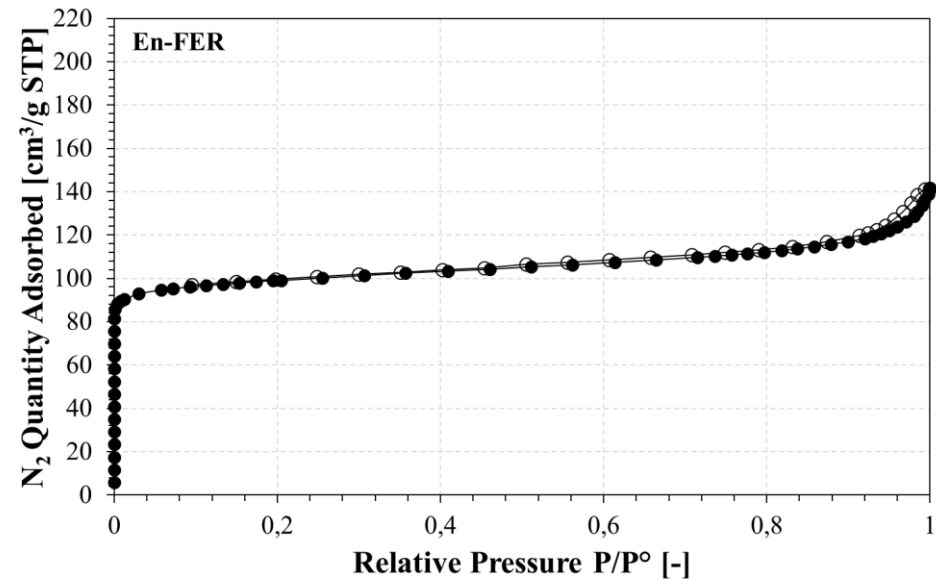
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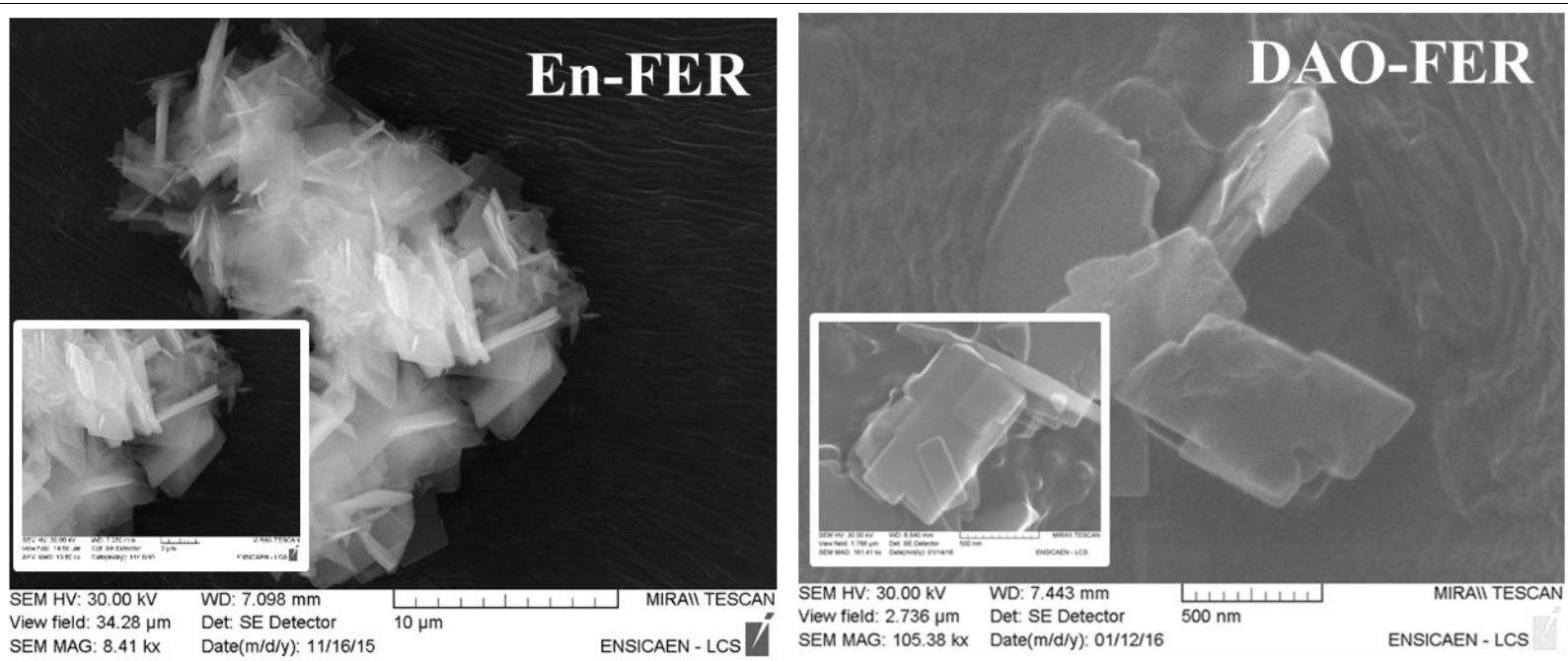
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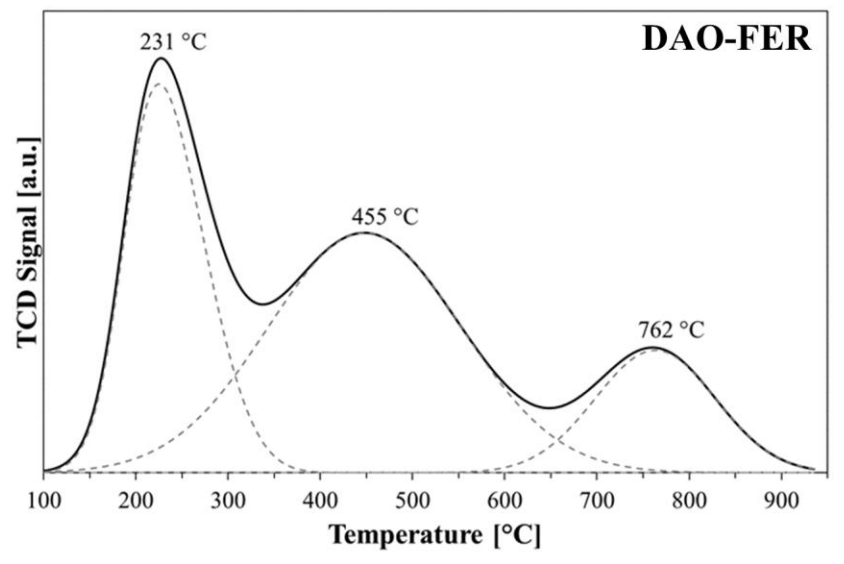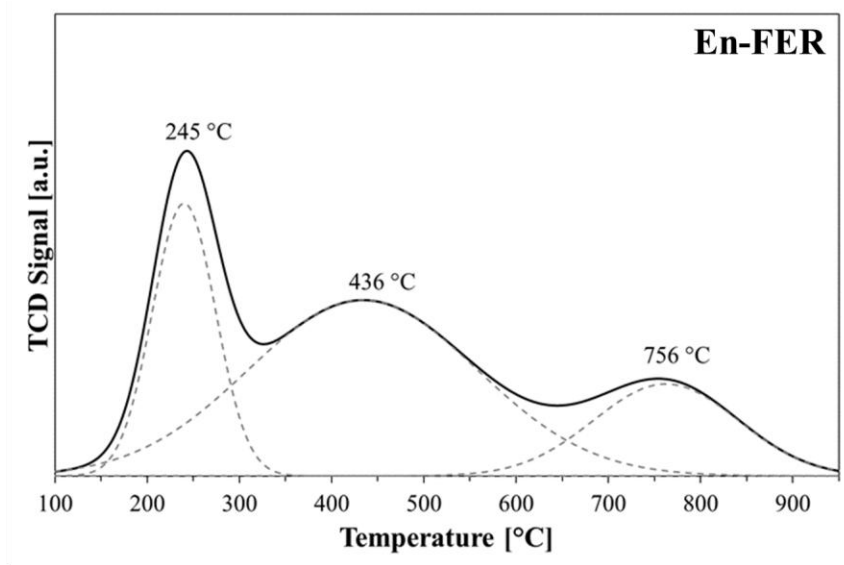
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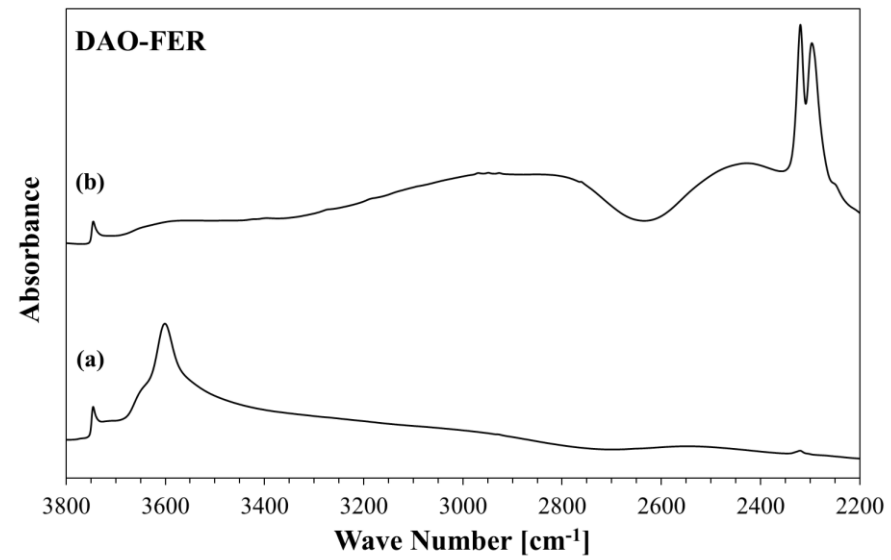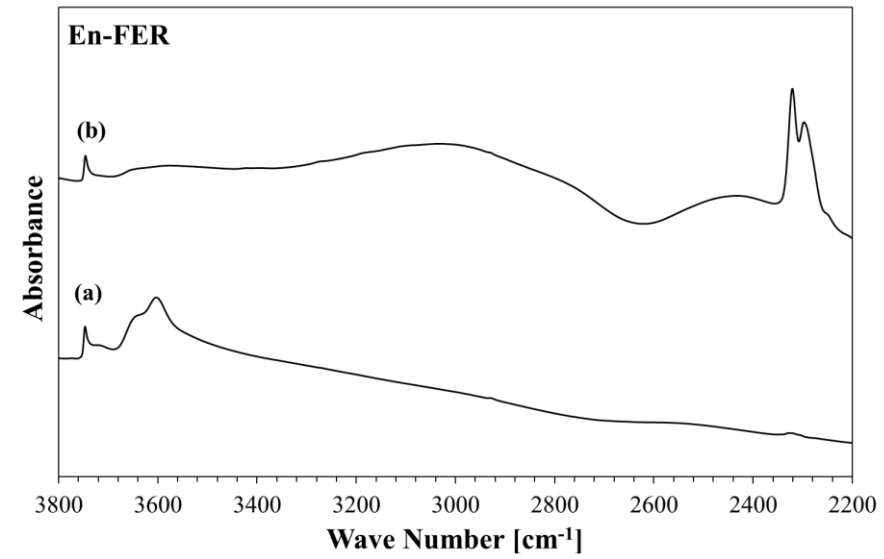
Figure 6

Figure 7

Figure 8