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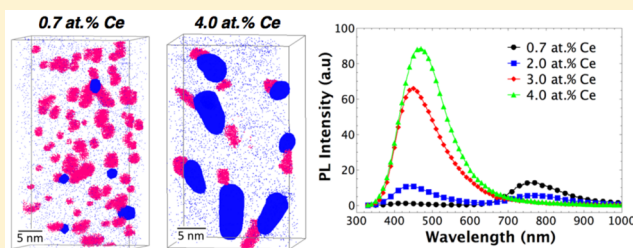
Direct Insight into Ce-Silicates/Si-Nanoclusters Snowman-Like Janus Nanoparticles Formation in Ce-Doped SiO_x Thin Layers

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ABSTRACT: The present work reports a nanoscale chemical and structural study on the influence of Ce content in Ce-doped SiO_{1.5} thin films via atom probe tomography (APT). Using this technique, we can explore 3D mapping of the atomic distribution inside a material. Such an investigation is crucial to optimize the optical properties of Ce-doped SiO_x films. As a result, we clearly identify the influence of cerium on the phase separation process, on the silicon nanocrystal growth, and on the formation of cerium silicate nanoparticles occurring during annealing treatments. The observed nanoscale structure is correlated with the optical properties measured by room temperature photoluminescence spectroscopy thus leading to a comprehensive understanding of the properties of Ce-doped SiO_{1.5} thin films.



INTRODUCTION

Silicon photonics has been benefiting from an unprecedented boom over the last two decades. Several keystone components have been successfully demonstrated with remarkable performance, such as extremely low-loss waveguides, high bandwidth photodetectors, and ultrafast modulators.¹ Despite these significant advances, the development of an efficient silicon-based light source has seriously lagged behind that of other photonic devices because of the low emission efficiency of bulk silicon.² To progress toward this aim, several works were dedicated to the integration of III–V semiconductors or other materials such as ZnO with silicon because of the direct and wide band gaps of these compounds.^{3–5} However, because of the lattice and thermal mismatch between III–V materials and silicon, the integration remains a challenging issue.⁶ Another promising approach, which has attracted a lot of interest, is doping of silicon oxide thin films (SiO_x) with rare-earth elements. Depending on the rare-earth element considered, the emission can be tuned from ultraviolet (UV) to infrared.^{7–12} Up to now, efficient visible-light-emitting devices based on rare-earth-doped metal-oxide-semiconductor (MOS) structures have been demonstrated.^{13,14} However, the solubility of rare-earth ions in Si-based materials achieved so far is still relatively low (10¹⁶–10²⁰ cm⁻³).¹⁵ Moreover, for high rare earth ion concentrations, deterioration of luminescence efficiency is observed because of formation of ion clusters or secondary phases (e.g., oxides or optically inactive silicates).^{16,17} Such a concentration quenching presents a substantial barrier for further performance improvement of rare-earth-doped Si-based materials. Recently, it was shown that formation of rare-earth silicates (e.g., Ce₂Si₂O₇, Eu₂SiO₅, Er₂Si₂O₇) can enhance their

photoluminescence efficiency.^{18–20} In these compounds, rare-earth ions are optically active, and their solubility can reach 10²² cm⁻³.^{20–22}

Among rare-earth silicates, Ce₂Si₂O₇ is a promising material for UV optoelectronic devices with an emission band close to that of ZnO and GaN. The photoluminescence properties of Ce₂Si₂O₇ have been investigated by several groups.^{18,19,23,24} Recently, we have reported the formation of Ce-silicate nanoclusters in substoichiometric 3 at. % Ce-doped SiO_{1.5} thin films. Their formation is the result of a complex phase separation at the nanoscale between Si atoms in excess, Ce atoms, and the SiO₂ matrix.²⁵ However, only one particular Ce concentration (i.e., 3 at. % Ce) was considered in this study. A systematic investigation of the effect of Ce concentration on the phase separation process could help to shed light into the complex nanoscale structural evolution of Ce-doped SiO_{1.5} thin films. In this paper, we perform an atomic scale investigation of Ce-doped SiO_{1.5} thin films as a function of Ce content by means of atom probe tomography. The observed nanoscale structure was then correlated with the optical properties studied by photoluminescence spectroscopy. The present paper leads to a comprehensive study of phase transformations occurring in Ce-doped SiO_{1.5} thin films and their relation with the optical properties.

EXPERIMENTAL SECTION

Sample Preparation. Substoichiometric $\text{SiO}_{1.5}$ thin films were deposited on top of $\text{Si}(001)$ substrates by coevaporation of SiO powder from a thermal cell and SiO_2 from an electron beam gun in an ultrahigh vacuum chamber. The film thickness was 200 nm for all the samples considered in this study. Ce-doping was performed from a Knudsen cell heated around 1400 °C. The Ce flux was monitored by a quartz microbalance, and the Ce content defined by $[\text{Ce}]/([\text{Si}] + [\text{O}] + [\text{Ce}])$ was varied between 0.7 at. % and 4 at. %. The samples were postgrown annealed in a rapid thermal annealing furnace under flowing N_2 for 5 min at 1100 °C. At this particular annealing temperature, a rather strong Ce-related photoluminescence was already observed.²⁵

Structural and Chemical Characterization. The 3D atomic scale characterization of the samples was performed using a laser-assisted wide-angle tomographic atom probe (LAWATAP-Cameca) instrument. Further information on atom probe tomography can be found elsewhere.²⁶ In an atom probe tomography experiment, specimens must be needle-shaped having an end radius below 50 nm. The preparation of such tips was achieved by employing lift-out and annular milling method, using a dual beam Zeiss NVision 40 focused ion beam scanning electron microscope (FIB-SEM).^{27,28} To prevent any damage during the cutting and milling process, samples were covered by a ~ 350 nm thick platinum layer before FIB processing. Photoluminescence (PL) measurements were performed at room temperature using the 325 nm line of a He–Cd laser for the sample excitation. The PL signal was analyzed by a monochromator equipped with a 150 grooves/mm grating and by a cooled charge coupled device camera with a 300–1050 nm detection range.

RESULTS AND DISCUSSION

Photoluminescence Measurements. To investigate the impact of the Ce content on the optical properties, PL measurements on $\text{SiO}_{1.5}$ films doped with a cerium concentration ranging from 0.7 to 4 at. % were performed. All samples were annealed at 1100 °C. The corresponding results are presented in Figure 1. Two emission bands are observed for Ce contents below 3 at. %. The first one corresponds to a blue emission band centered at 450 nm. The

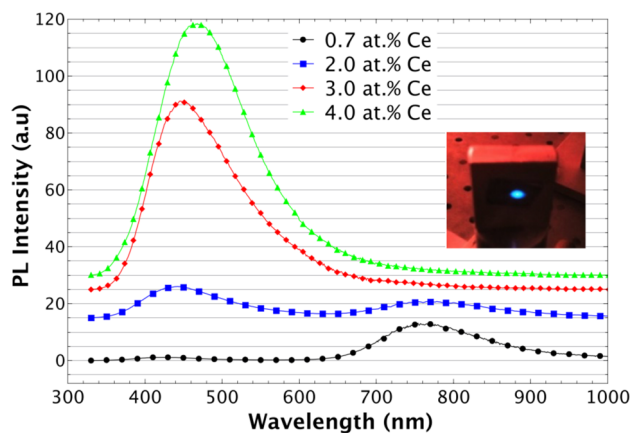


Figure 1. Room temperature (RT) PL spectra of Ce-doped $\text{SiO}_{1.5}$ thin films annealed for 5 min at 1100 °C. The Ce concentration is varied from 0.7 at. % to 4 at. %. The inset shows the blue luminescence visible at RT with the naked eyes.

emission can be attributed to the electric dipolar allowed $5d \rightarrow 4f$ transition of Ce^{3+} ions. The second one, in the near-infrared centered at 750 nm, is due to radiative recombinations between electrons and holes confined in silicon nanoclusters (Si-NCs).²⁹

The existence of Si-NCs in the layers and their optical signature is expected. Indeed, it is well-known that the presence of Si atoms in excess in substoichiometric SiO_x thin films can result in the formation of Si nanocrystals when the films are annealed at sufficiently high temperatures.³⁰ When the Ce content in the films increases, the Si-NCs related luminescence decreases while simultaneously that of Ce increases, suggesting an influence of the Ce on the growth and on the optical properties of Si-NCs. For the 0.7 and 2 at. % Ce-doped samples, the Ce^{3+} related luminescence is weak. Yet, a significant improvement was observed for the 3 and 4 at. % Ce-doped layers, indicating that the Ce-related luminescence does not increase linearly with the Ce concentration. The Ce-related luminescence from the 3 at. % and 4 at. % doped samples is strong and can be even seen with the naked eyes (see inset of Figure 1). Moreover, we can note a red-shift of the Ce-related luminescence when the Ce concentration increases to 3 and 4%. This may possibly be due to a modification of the local environment of Ce^{3+} ions. To elucidate the mechanism underlying the evolution of PL intensities and the quenching of the Si-NCs related PL for high Ce contents (between 2 and 3 at. %), the nanoscale structure of the samples was investigated by atom probe tomography.

Structural Evolution. All the samples were analyzed by atom probe tomography. Figure 2 shows the obtained 3D maps for which cluster identification was performed on Si and Ce to detect the presence or absence of precipitates in the samples using the GPM's data treatment code. In all samples, pure Si nanoclusters (pink) and Ce-rich clusters (blue) were identified. Moreover, we observe that Ce-rich clusters are always localized at the vicinity of a Si-NC whatever the Ce concentration. The sample doped with 0.7 at. % of Ce presents a high number density of Si nanoparticles (Figure 2a) which are well separated from Ce nanoclusters. For the 2 at. % Ce doped layer (Figure 2b), the number density of small, isolated Si-NCs strongly decreases. Meanwhile, the population of Ce-rich clusters, as well as that of large, coalesced Si nanoparticles increases. Further increase in the Ce content up to 3 at. % and 4 at. % (Figure 2c and 2d, respectively) leads to an additional decrease in the number density of Si nanoparticles. Furthermore, all Si nanoparticles identified in the volume for 3 and 4 at. % of Ce are linked to Ce-rich nanoclusters. The impact of Ce content on the growth of Si nanoparticles is clearly demonstrated. The 3D reconstructions reveal the existence of two populations of Si nanoparticles: small ones, isolated in the matrix, or larger ones bounded to Ce-rich nanoclusters. The isolated Si-NCs appear only for low Ce contents (0.7 and 2 at. %), whereas, for the 3 and 4 at. % Ce-doped layers, all the silicon nanoparticles are located in the immediate proximity of Ce-rich precipitates. The modification of the Si nanoparticle growth mechanism in the layers is strongly correlated with the increase of the Ce content. As it can be seen in Figure 2, their size and number density are clearly strongly impacted. Subsequently, we will look to the quantification of these parameters in order to better understand the mechanisms governing this process. The nanoparticles are spherical, and the observed elongated shape is due to local magnification effects during APT experiment analysis as previously demonstrated on similar systems.^{25,31}

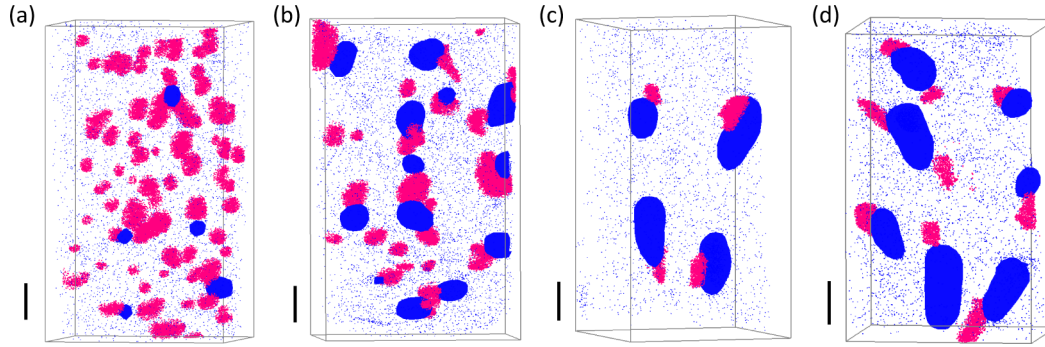


Figure 2. Distribution of Si nanoparticles (pink) and Ce-rich clusters (blue) in the analyzed volumes obtained by atom probe tomography for Ce-doped $\text{SiO}_{1.5}$ thin films: (a) 0.7 at. % Ce, (b) 2 at. % Ce, (c) 3 at. % Ce, and (d) 4 at. % Ce. Isolated blue dots correspond to free Ce atoms in the SiO_2 matrix. Scale bar is 5 nm.

Composition Measurements. First, we will focus on the identification of the phases existing in the samples. In all the samples, the matrix composition corresponds to SiO_2 comprising each time in addition a fraction of isolated Ce atoms as observed in Figure 2 (identified by isolated blue dots). All other Ce ions are located in the Ce-rich particles. The average compositions of the Ce-rich particles are presented on the ternary O–Ce–Si phase diagram shown in Figure 3. For

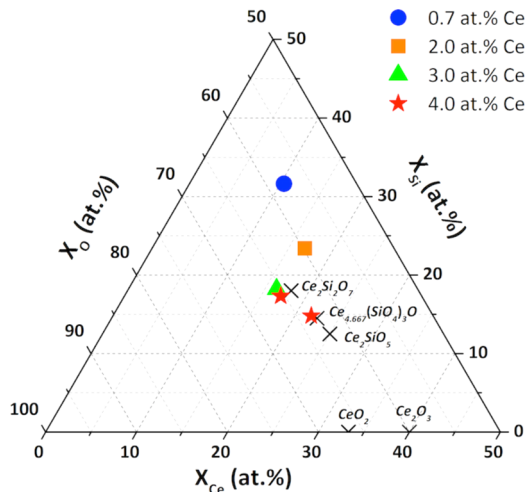


Figure 3. Mean composition measurements of Ce-rich clusters determined from APT analysis for the different Ce concentrations plotted on the ternary O–Ce–Si phase diagram.

the 0.7 and 2 at. % Ce-doped samples, the obtained composition does not match with any stoichiometry of a defined cerium silicate compound. A stable cerium silicate phase of stoichiometry $\text{Ce}_2\text{Si}_2\text{O}_7$ is identified for the 3 at. % Ce-doped sample. Finally, for the 4 at. % Ce-doped sample, the measurements give evidence of the existence of two compounds with different stoichiometry: $\text{Ce}_2\text{Si}_2\text{O}_7$ and $\text{Ce}_{4.667}(\text{SiO}_4)_3\text{O}$. Their amount in the studied volume corresponds to 43% and 57%, respectively, for $\text{Ce}_2\text{Si}_2\text{O}_7$ and $\text{Ce}_{4.667}(\text{SiO}_4)_3\text{O}$. The coexistence of these two compounds has already been mentioned in the literature.^{19,24} High-temperature annealing of cerium oxide layers leads to the formation of cerium silicates at the interface with the silica or silicon substrates. In this contribution, we evidenced the formation of cerium silicates in the form of nanoparticles in substoichiometric Ce-doped $\text{SiO}_{1.5}$ films. Cerium silicates can exist in

various polymorphs.³² The identification of such polymorphs was, however, not possible and appears to be clearly beyond the scope of the present paper. As mentioned previously, all Ce atoms did not precipitate into clusters, and a small quantity remains diluted in the silica matrix. This quantity allows us to determine the solubility limit of Ce in silica or silicon-rich silicon oxide films assuming that coalescence stage is reached. On the basis of the measurement of the Ce atomic fraction in the silica matrix, the solubility limit of Ce in SiO_2 can be estimated to be $0.8 \pm 0.2\%$ (i.e., $5.3 \pm 1.3 \times 10^{20} \text{ cm}^{-3}$) for samples annealed at 1100 °C. Finally, all Si atoms in excess have precipitated into pure Si nanoparticles or are located in the Ce-rich clusters.

Morphological Information. To quantitatively evaluate the influence of Ce on the growth of silicon nanoparticles, we have measured the number density and the radius of both Ce nanoclusters and Si-NCs (isolated or linked to Ce nanoclusters).

Figure 4a shows the evolution of the density of nanoparticles with the nominal Ce concentration in the films. The number density of the isolated Si-NCs monotonously decreases with the Ce content until 3%. The evolution of the number density of linked Si-nanoparticles (shown as triangle in Figure 4a) reveals that this value is nearly constant indicating a strong decrease of the number density of isolated Si-NCs. This highlights the effect of Ce on the growth mechanism of Si nanoparticles that does not take place as in undoped SiO_x films. Indeed, as the number of Si atoms in excess and the annealing temperature are constant, the density of Si-NCs should be constant if the Ce atoms do not influence the Si-NCs growth mechanism. The Ce-rich clusters number density increases substantially as the Ce content increases from 0.7 to 2 at. %. This evolution can be attributed to the increase of the Ce concentration which induces more inhomogeneity in the sample and germination sites but without formation of any stable compound. However, for the 3 at. % and 4 at. % doped samples, the formation of cerium silicates induces a decrease of the Ce-rich particle number density.

Figure 4b shows the evolution of the mean diameter of both Ce-rich and Si nanoclusters with the Ce content in the films. In this figure, we divide the concentration effect in three regimes: regime (I) corresponding to Ce concentrations up to 2 at. %, where isolated Si-NCs are present, regime (II) corresponding to Ce contents between 2 and 3 at. %, and finally, regime (III) corresponding to Ce contents beyond 3 at. % where only linked Si-NCs can be observed.

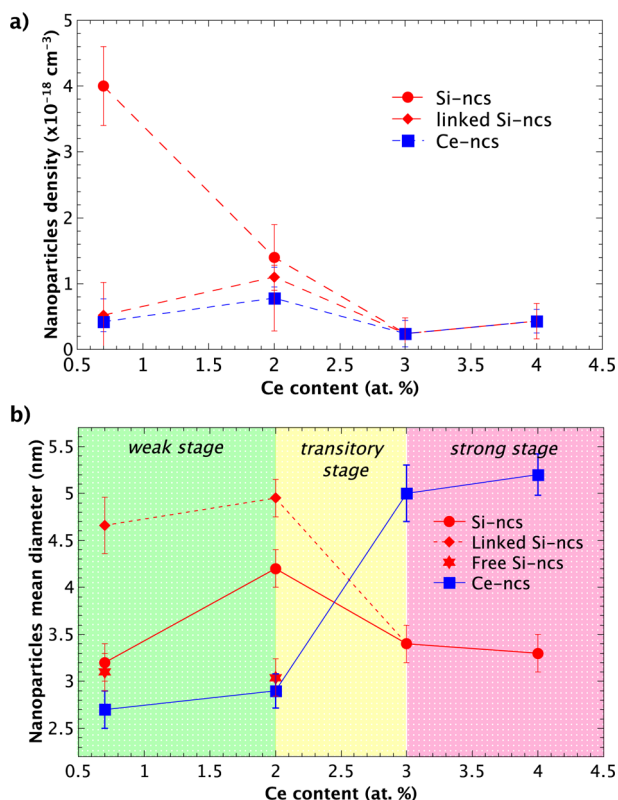


Figure 4. (a) Density of nanoparticles (silicon or Ce-rich) as a function of the Ce content. (b) Mean diameter of nanoparticles (silicon or Ce-rich) as a function of the Ce content.

The mean size of the Ce-rich clusters does not change in either regime I or III. However, their size increases by nearly a factor of 2 when the Ce content increases from 2 at. % to 3 at. %. This evolution may be related to an increasing consumption of Ce atoms (as evidenced by the increase of the mean Ce composition of Ce-rich nanoclusters in Figure 3 and by the decrease of the Ce-rich nanocluster density shown in Figure 4a) leading eventually to the formation of cerium silicate nanoparticles.

Regarding the diameter of Si nanoparticles (Figure 4b), the average size measured for the sample containing 0.7 at. % Ce is 3.2 ± 0.2 nm. For this sample, 90% of Si-NCs are isolated in the matrix. These free particles have an average size of 3.1 ± 0.2 nm. In contrast, the other 10% of Si-NCs (linked to Ce-NCs) have an average size of 4.6 ± 0.3 nm, which is larger than the size of the free particles. For a concentration of 2 at. % Ce, we observe an increase in the Si-NCs average size. The mean size of isolated Si-NCs is 3.0 ± 0.2 nm. This value is comparable with that measured for the sample doped with 0.7 at. % Ce in which 90% of Si-NCs were free. Thus, the free particles are not affected by the presence of Ce. Linked Si nanoparticles have a mean size of 4.9 ± 0.2 nm. The population of linked Si-NCs increases from 10% for the 0.7 at. % Ce-doped sample to 55% for the 2 at. % Ce-doped sample. This increase in the proportion of linked Si-NCs is therefore responsible for the increase in the average size of the sample doped with 2 at. % Ce. In samples doped with 3 at. % Ce and 4 at. % Ce, all Si-NCs are linked to cerium silicate nanoparticles, and their average size decreases to 3.4 ± 0.2 nm. As the Ce content increases, an increasing number of stoichiometric cerium silicate nanoparticles are formed thereby consuming Si atoms in excess. As a result, less Si atoms are available to form pure Si-NCs. This can be explained by the fact that with the increase of the amount of Ce, stoichiometric cerium silicates are formed which consume more Si atoms, and less Si atoms are ejected during the phase separation, and therefore, less Si excess is available to form Si-NCs.^{25,33} For high Ce concentrations (3 at. %), areas rich in Ce, O, and Si are initially formed during the deposition process. Then, Si nanoparticles are created through a phase separation mechanism from Ce–Si–O rich particles to Si-NCs and cerium silicate. Therefore, no classical germination growth of free Si-NCs is observed up to 2 at. % of Ce.

Precipitation Mechanism. To explain the growth of cerium silicate nanoparticles and the evolution of linked versus free Si-NCs, we propose a precipitation mechanism as a function of annealing temperature and Ce content. We have previously reported a phase separation occurring at the nanoscale between pure Si-NCs and cerium silicate NCs in a $\text{SiO}_{1.5}$ film doped with 3 at. % Ce.²⁵

The diagram shown in Figure 5 summarizes the morphological and chemical changes observed by atom probe

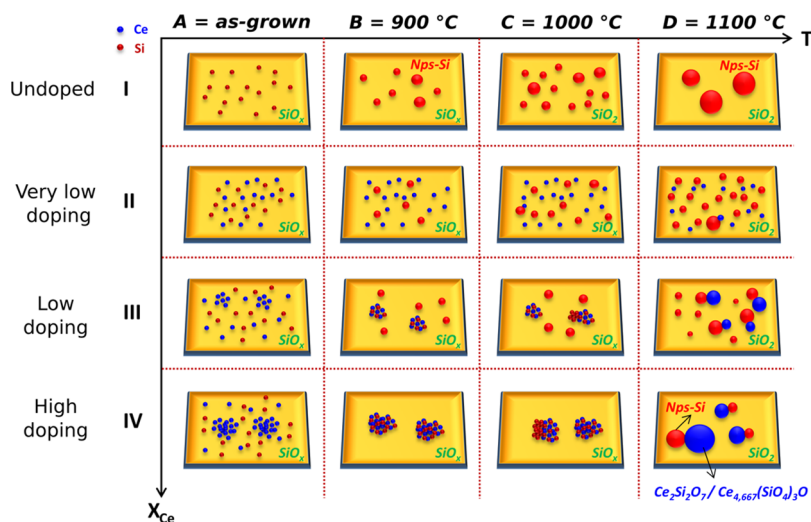


Figure 5. Schematic representation of the precipitation mechanism in Ce-doped $\text{SiO}_{1.5}$ thin films as a function of temperature and Ce doping level.

tomography experiments performed on Ce-doped $\text{SiO}_{1.5}$ thin films as a function of annealing temperature and Ce content. In this diagram, we have divided the influence of Ce content on the diffusion and precipitation of Si and cerium silicates with annealing temperature in four different regimes.

Regime I. Undoped SiO_x Films. For the as-deposited sample (I.A), silicon atoms can be either distributed homogeneously or not, depending on the incorporated Si excess.^{34,35} The thermal annealing activates the phase separation between Si and SiO_2 . During the early stages of annealing (I.B), germination and growth of pure Si nanoparticles occur, and the matrix still contains Si atoms in excess. During the precipitation process (I.C), the excess of Si is consumed gradually until that oversaturation is completely consumed. When all Si atoms in excess are precipitated, the precipitation process is complete, and the matrix is composed of SiO_2 . Moreover, the growth rate and diffusion coefficient strongly depends on the elaboration conditions and the amount of Si atoms in excess. In the last stage (I.D), coalescence takes place in the films leading to an increased particle size and to a reduced density.

Regime II. Low-Doped Cerium Films ($X_{\text{Ce}} < \text{Ce Solubility Limit in } \text{SiO}_2$). The Ce atoms do not significantly affect the precipitation of the Si nanoparticles. Indeed, in the case of the 0.7 at. % Ce-doped film, 90% of the Si nanoparticles remain isolated. Moreover, their mean size and density are compatible with those measured on undoped SiO_x films. Whatever the annealing temperature, the formation of Si nanoparticles is almost identical to that of regime I, and the majority of cerium ions remains diluted in the matrix. In this case, segregation of Ce is not observed.

Regime III. Intermediate-Doped Cerium Films (Ce Solubility Limit in $\text{SiO}_2 < X_{\text{Ce}} < \sim 2$ at. %). The formation of small Ce seeds is observed while the distribution of Si atoms remains homogeneous in as-grown films (IIIA).²⁵ During annealing, Si atoms diffuse toward the Ce seeds to form nonstable Si–O–Ce rich clusters. Meanwhile, the diffusing Si atoms also lead to the formation of Si nanoparticles in a conventional manner (IIIB). Two mechanisms thus occur simultaneously: a phase separation between Si and SiO_2 and an accumulation of Ce atoms exceeding locally the segregation of cerium in excess compared to the solubility limit. In this regime, the Ce content in the Si–O–Ce rich clusters is still too low to form a stoichiometric cerium silicate phase. There is, therefore, a creation of local inhomogeneity of Ce with elaboration and slight change in the first stages of annealing. On the other hand, SiO_x phase separation tends to eject Si from SiO_2 . The Si atoms in excess enter either into pure Si nanoparticles or into Si–O–Ce clusters. On the basis of our observations, it appears that Si atoms in excess located close to Ce-rich clusters ejected Si that forms silicon nanoparticles and areas rich in Si–Ce–O. Silicon in excess near Ce-rich zones go to those areas rather than forming isolated and pure Si nanoparticles.

Regime IV. Case of Highly Doped Cerium Films ($X_{\text{Ce}} > 3$ at. %). In this regime, the distribution of Ce atoms is increasingly inhomogeneous while the distribution of Si atoms remains homogeneous in as-grown films. During annealing, the Si atoms in excess will mainly diffuse toward the Ce-rich seeds (IV.B). For annealing temperatures larger than 900 °C, Si atoms start to be ejected from the Si–Ce–O clusters, and eventually, pure Si-nanocrystals start to form.²⁵ For annealing temperatures close to 1100 °C, the thermal budget allows the precipitation of cerium silicate nanoparticles to minimize the free energy of the system. The Si–Ce–O clusters thus completely dissociate into

pure Si-NCs, and cerium silicate nanoparticles thus lead to the observation of snowman-like nanoparticles. The system tends to form a cerium silicate phase to minimize its energy. With additional thermal contributions, the system energy is changed by increasing the degree of disorder or entropy. Once the matrix consisting of $\text{SiO}_2 + \text{Ce}$ is diluted, a phase separation at the nanoscale is thus observed between Si and Si–Ce–O (IV.C). Si atoms are ejected from these areas. At higher temperature, the energy supply is sufficient to complete this phase separation. Rich areas dissociate completely into pure Si nanoparticles and cerium silicate phases. The lack of thermodynamic data on cerium silicates limits the explanation of the observed reaction and, therefore, which silicate is the most energetically favorable phase or not. The observed nanostructure is remarkable. Si nanoparticles grow on the surface of cerium silicates. This nanostructure which has been identified in the literature as snowman-like nanoparticles was mainly observed during synthesis of colloidal nanoparticles.^{36,37} To our knowledge, the demonstration of such a configuration in thin films has never been reported.

The proposed precipitation mechanism has been extracted by varying the Ce content at a constant silicon excess of 10%. On the basis of our previous work on Er-doped silicon-rich silica with less silicon excess,¹⁷ we observed a similar formation of both pure Si-NCs and Er-rich nanoparticles, highlighting that the precipitation mechanism can be generalized to other RE^{3+} ions. However, this assumption should be confirmed by a systematic investigation of highly doped nanostructures and for high-annealing temperatures. On the other hand, the influence of silicon excess on the phase separation process can be divided into two stages: (a) first, at low annealing temperature and depending on the RE dopant content, a portion or all of the silicon atoms in excess are consumed to form RE–Si–O rich clusters. (b) Second, at high-annealing temperature, the RE–Si–O rich clusters will be decomposed into stable RE-silicate phases and pure Si-NCs.²⁵ Therefore, the density and the composition of the formed nanoparticles are highly influenced by the silicon excess and the RE content.

Correlation between the Nanostructure and the Optical Properties. The nanostructures observed previously by atom probe tomography help us to understand the changes in the optical properties shown in Figure 1. Indeed, the weak Ce-related luminescence observed for the samples doped with 0.7 at. % and 2 at. % Ce originates from isolated Ce^{3+} ions. Indeed, for the sample doped with 0.7 at. % Ce showing the lowest Ce-related luminescence, most of the Ce ions (95%) remain isolated in the matrix. For the sample doped with 2 at. % Ce, the Ce-related luminescence increases and further slightly shifts to larger wavelengths. This indicates that some Ce ions must have a different chemical environment. Indeed, we found that a small amount of cerium silicate nanoparticles are already present in the sample doped with 2 at. % Ce in addition to the isolated Ce^{3+} ions. This can be explained by the precipitation and the formation of a small amount of Ce silicate ($\text{Ce}_2\text{Si}_2\text{O}_7$) (20%). For higher Ce contents, the Ce-related luminescence increases dramatically while a clear shift is observed to higher wavelengths indicating that an increasing number of Ce atoms participate in the formation of optically active cerium silicates nanoparticles. In contrast, the Si-NCs related luminescence decreases gradually as the Ce content increases in the films. This may possibly be related to an increasing size of the Si-NCs which could lead to a loss of the quantum confinement. However, we have shown that the Si-

NC size remains well below 5 nm (Figure 4b) thus allowing the exclusion of this first hypothesis. We thus explain the observed evolution by the decreasing density of free Si-NCs (Figure 4a) for high Ce contents. For the samples doped with 3 at. % and 4 at. % Ce, all Si-NCs are indeed linked to the cerium silicate nanoparticles (Figure 2). We expect that the close vicinity of cerium silicate nanoparticles will provide nonradiative recombination pathways for excitons confined in pure Si-NCs.

CONCLUSIONS

To summarize, our investigations helped to shed light into the complex pathways leading to precipitation and phase separation in Ce-doped SiO_{1.5} thin films. Thanks to atom probe tomography, which allows a 3D reconstruction of the matter at the atomic scale, we provide evidence of the formation of snowman-like nanostructures consisting of pure Si-NCs linked to cerium silicate nanoparticles. The influence of Ce content on the formation of Si-NCs and cerium silicate nanoparticles is discussed. On the basis of our observations, we were able to propose a precipitation mechanism in Ce-doped SiO_{1.5} thin films as a function of both annealing temperature and Ce content. This allowed us to get a direct comprehensive view of the optical properties of the samples investigated by room-temperature photoluminescence. Such a deep correlation between optical properties and structure at the nanoscale represents an important step toward the realization of future Si-based light-emitting devices.

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